

University of Groningen

Dirac Nodal Arc Semimetal PtSn₄

Li, Guowei; Fu, Chenguang; Shi, Wujun; Jiao, Lin; Wu, Jiquan; Yang, Qun; Saha, Rana; Kamminga, Machteld E.; Srivastava, Abhay K.; Liu, Enke

Published in:
Angewandte Chemie (International ed. in English)

DOI:
[10.1002/anie.201906109](https://doi.org/10.1002/anie.201906109)

IMPORTANT NOTE: You are advised to consult the publisher's version (publisher's PDF) if you wish to cite from it. Please check the document version below.

Document Version
Publisher's PDF, also known as Version of record

Publication date:
2019

[Link to publication in University of Groningen/UMCG research database](#)

Citation for published version (APA):

Li, G., Fu, C., Shi, W., Jiao, L., Wu, J., Yang, Q., Saha, R., Kamminga, M. E., Srivastava, A. K., Liu, E., Yazdani, A. N., Kumar, N., Zhang, J., Blake, G. R., Liu, X., Fahlman, M., Wirth, S., Auffermann, G., Gooth, J., ... Felser, C. (2019). Dirac Nodal Arc Semimetal PtSn₄: An Ideal Platform for Understanding Surface Properties and Catalysis for Hydrogen Evolution. *Angewandte Chemie (International ed. in English)*, 58(37), 13107-13112. <https://doi.org/10.1002/anie.201906109>

Copyright

Other than for strictly personal use, it is not permitted to download or to forward/distribute the text or part of it without the consent of the author(s) and/or copyright holder(s), unless the work is under an open content license (like Creative Commons).

The publication may also be distributed here under the terms of Article 25fa of the Dutch Copyright Act, indicated by the "Taverne" license. More information can be found on the University of Groningen website: <https://www.rug.nl/library/open-access/self-archiving-pure/taverne-amendment>.

Take-down policy

If you believe that this document breaches copyright please contact us providing details, and we will remove access to the work immediately and investigate your claim.

Downloaded from the University of Groningen/UMCG research database (Pure): <http://www.rug.nl/research/portal>. For technical reasons the number of authors shown on this cover page is limited to 10 maximum.

Dirac Nodal Arc Semimetal PtSn₄: An Ideal Platform for Understanding Surface Properties and Catalysis for Hydrogen Evolution

Guowei Li^{+,*}, Chenguang Fu⁺, Wujun Shi⁺, Lin Jiao, Jiquan Wu, Qun Yang, Rana Saha, Machteld E. Kamminga, Abhay K. Srivastava, Enke Liu, Aliza N. Yazdani, Nitesh Kumar, Jian Zhang, Graeme R. Blake, Xianjie Liu, Mats Fahlman, Steffen Wirth, Gudrun Auffermann, Johannes Gooth, Stuart Parkin, Vidya Madhavan, Xinliang Feng, Yan Sun^{+,*} and Claudia Felser^{+,*}

Abstract: Conductivity, carrier mobility, and a suitable Gibbs free energy are important criteria that determine the performance of catalysts for a hydrogen evolution reaction (HER). However, it is a challenge to combine these factors into a single compound. Herein, we discover a superior electrocatalyst for a HER in the recently identified Dirac nodal arc semimetal PtSn₄. The determined turnover frequency (TOF) for each active site of PtSn₄ is 1.54 H₂s⁻¹ at 100 mV. This sets a benchmark for HER catalysis on Pt-based noble metals and earth-abundant metal catalysts. We make use of the robust surface states of PtSn₄ as their electrons can be transferred to the adsorbed hydrogen atoms in the catalytic process more efficiently. In addition, PtSn₄ displays excellent chemical and electrochemical stabilities after long-term exposure in air and long-time HER stability tests.

Introduction

The hydrogen evolution reaction (2H⁺ + 2e⁻ → H₂, HER) that occurs at the solid–liquid interface plays a vital role in clean energy conversion and for the understanding of other complicated heterogeneous reactions. The use of suitable catalysts that have high efficiency, selectivity, and

stability is crucial for practical applications.^[1] Many advanced electrocatalysts have been designed and developed by employing a variety of strategies.^[2] However, a suitable lower cost replacement of the benchmark Pt catalyst has not yet been developed. The dominance of Pt as the best catalyst is an inevitable result of several key characteristics. 1) The excellent chemical stability of Pt provides active sites for obtaining a high turnover frequency (TOF). 2) Its ultra-high electrical conductivity guarantees fast electron transport and reduces the Schottky barrier at the interface between the Pt and adsorbates. 3) A suitable d-band center favors the metal-adsorbate interaction and hydrogen release process.^[3]

Recent observations of unoccupied metallic topological surface states (TSSs) suggest that certain noble metals, such as Pt are in fact Z₂ topological metals.^[4] The metallic surface states of these metals are protected by time reversal symmetry and are robust against surface modification and defects under ambient conditions.^[5] These are precisely the conditions that are beneficial for the various surface reactions that occur on the crystal surface.^[6] Our previous study on a range of topological materials from bismuth chalcogenide topological insulators^[7] to Weyl semimetals,^[8] demonstrated that each of these could be promising candidates for use as catalysts.

[*] Dr. G. Li,^[+] Dr. C. Fu,^[+] Dr. L. Jiao, Q. Yang, Dr. E. Liu, N. Kumar, S. Wirth, G. Auffermann, Dr. J. Gooth, Dr. Y. Sun,^[+] Prof. C. Felser^[+]
Max Planck Institute for Chemical Physics of Solids
01187 Dresden (Germany)
E-mail: Guowei.Li@cpfs.mpg.de
Yan.Sun@cpfs.mpg.de
Claudia.Felser@cpfs.mpg.de

Dr. W. Shi^[+]
School of Physical Science and Technology, ShanghaiTech University
201203 Shanghai (China)

Dr. J. Wu, Dr. X. Liu, Prof. M. Fahlman
Department of Physics, Chemistry and Biology (IFM), Linköping University
58183 Linköping (Sweden)

Dr. R. Saha, Dr. A. K. Srivastava, Prof. S. Parkin
Max Planck Institute for Microstructure Physics
06120 Halle (Germany)

Dr. M. E. Kamminga, Dr. G. R. Blake
Zernike Institute for Advanced Materials, University of Groningen
9747 AG Groningen (The Netherlands)

A. N. Yazdani
Department of Chemistry, Carleton College
MN 55057 Northfield (USA)

Dr. J. Zhang, Prof. X. Feng
Department of Chemistry and Food Chemistry, Technische Universität Dresden
01062 Dresden (Germany)

Prof. V. Madhavan
Department of Physics and Frederick Seitz Materials Research Laboratory, University of Illinois Urbana-Champaign
Urbana, Illinois 61801 (USA)

[*] These authors contributed equally to this work.

Supporting information and the ORCID identification number(s) for the author(s) of this article can be found under:
<https://doi.org/10.1002/anie.201906109>.

© 2019 The Authors. Published by Wiley-VCH Verlag GmbH & Co. KGaA. This is an open access article under the terms of the Creative Commons Attribution License, which permits use, distribution and reproduction in any medium, provided the original work is properly cited.

However, both topological insulators and Dirac/Weyl semimetals exhibit very low carrier densities around the Fermi level owing to weak electrostatic screening. We approach this problem by considering a topological nodal arc. In the case of Dirac node arc semimetals, the band crossing points around the Fermi level form a Dirac node arc in momentum space. Many exotic properties, such as ultrahigh magnetoresistivity, carrier mobility are related to the Dirac fermions and topological surface states.^[9] This suggests the following question: can the Pt atomic layer be combined with the exotic physical properties related to the Dirac node arc to speed up the HER process?

To pursue this question, we examined the bulk single crystal Dirac semimetal PtSn₄. Recently, a high density of conduction electrons and Dirac node arcs were observed in single-crystalline PtSn₄ and subsequently confirmed using de Haas-van Alphen quantum oscillations.^[9c,10] The extremely weak bonding between the Sn-Pt-Sn layers of the compound enables the exposure of the Pt skin without the presence of metastable dangling bond surface states. We observe an ultra-low resistivity of only 42 $\mu\Omega\text{cm}$ and a high mobility of 200 $\text{cm}^2\text{V}^{-1}\text{s}^{-1}$ at room temperature in bulk single-crystalline PtSn₄, where the robust surface states are mainly derived from the p orbitals of Pt skin atoms. These lend a considerable contribution to the local density of states (DOS) near the Fermi Level and could act as electron donors for adsorbed molecules. With the bulk single crystal acting as both the cathode and catalyst for the HER, Pt-like activity is exhibited together with a decrease in Pt usage of 71 % in mass. The TOF value for the PtSn₄ bulk single crystals reached 1.54 H_2s^{-1} at 100 mV, surpassing those of state-of-the-art Pt/C, MoS₂, and other noble-metal based catalysts. Of greater importance is the superior electrocatalytic stability of the single crystal after a long-time and high current density test. These observations may be used to develop a new design principle for high intrinsic electrochemical catalysts with defined active sites.

Results and Discussion

Starting with Pt, we checked all of the Pt_xM_y (*M* is a main group element) binary compounds, and selectively short-listed specific representative cases that has the lowest Pt content with a certain *M* element. As shown in Figure 1a, PtSn₄ is chosen for this study because of its facile large-scale synthesis and high electrical conductivity. As a widely accepted descriptor in prediction HER activity, the Gibbs free-energy of the intermediate state, $|\Delta G_{\text{H}^*}|$, was calculated for different models (Figure S1 in the Supporting Information).^[11] A Pt55 metal cluster was also included for comparison.^[12] As shown in the Volcano plot (Figure 1b), Pt exhibits the optimal level for H adsorption with a near zero value of $\Delta G_{\text{H}^*} = -0.11$ eV, suggesting the best electrocatalytic activity from a thermodynamic viewpoint. In stark contrast, the exposure of the Pt layer in the PtSn₄ (010) plane resulted in

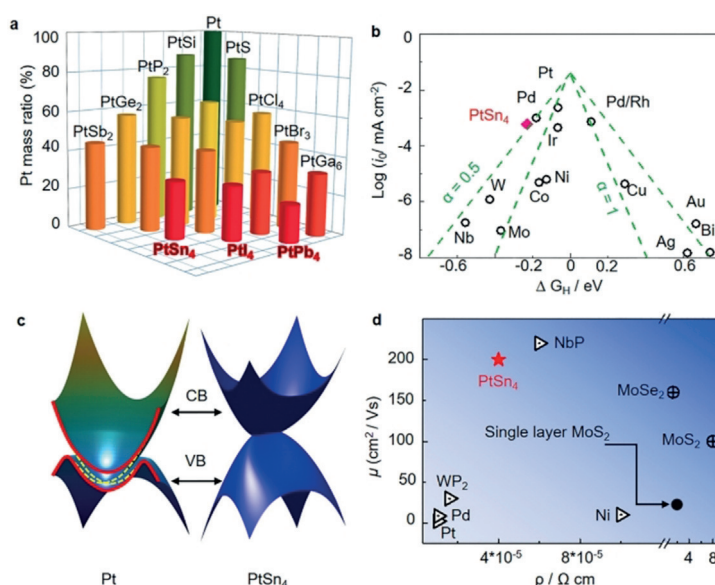


Figure 1. Motivations for searching for advanced electrocatalysts beyond pure Pt: a) The Pt mass ratios in binary compounds that include Pt. b) HER volcano plot for PtSn₄ and selected metal catalysts. The Gibbs free energy of PtSn₄ is very close to that of pure Pt. c) Band structure of Pt. The metallic topological surface states are shown in yellow. The bulk states (upper and lower parts) and surface states (middle part) are separated by the red line. The band structure of PtSn₄ with non-closed Dirac node arc is also shown. d) The room temperature mobility and conductivity of PtSn₄ with the selected advanced electrocatalysts.

a ΔG_{H^*} value of -0.28 eV, which is better than that of the Pt55 clusters, with a value of -0.33 eV. These results indicate that the Pt atomic layer of the PtSn₄ single crystal could be an ideal surface for HERs.

Figure 1c shows the band structure of Pt, and the topological surface states above the Fermi energy on the (111) surfaces owing to the bulk Z_2 topological order can be observed. Although the Z_2 topological order does not exist in PtSn₄, it hosts a Dirac nodal arc in the bulk band structure (Figure 1c).^[9c] This results in a large density of states around the Fermi level. Dirac electrons exist in the sheet, which generally correspond to a large carrier mobility. Figure 1d shows a comparison of the conductivity and mobility of a selection of advanced HER catalysts, which indicates the advantages of the PtSn₄ single crystal. The mobilities of the electrons and holes are very similar between 5 and 300 K and reach 200 $\text{cm}^2\text{V}^{-1}\text{s}^{-1}$ at room temperature, which is even higher than that of back-gated MoS₂, MoSe₂, WS₂, WSe₂, and black phosphorus on SiO₂ substrates.^[13]

With the above theoretical investigation and experimental facts in mind, high quality PtSn₄ single crystals were grown out of an Sn-rich binary melt. The HER performance evaluation of this single crystal was carried out in a conventional three-electrode electrochemical cell in 1M KOH electrolyte (the details are provided in the Methods section, Figure S2). The obtained polarization curves are shown in Figure 2a. A sharp increase in the cathode current with an onset overpotential of approximately 0 mV was observed on the PtSn₄ single crystal, with a burst of gas bubbles that appeared on the crystal surface (Figure S3 and Movie S1). To achieve a current density of 10 mA cm^{-2} , a low overpotential

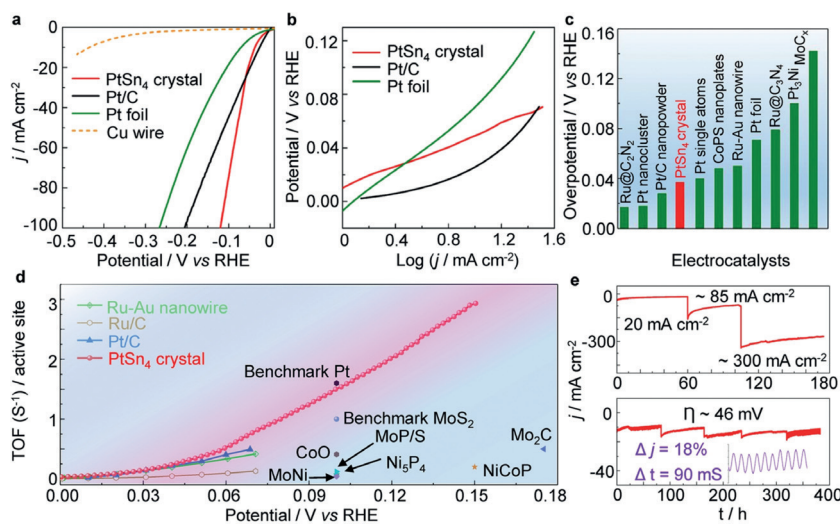


Figure 2. Electrochemical performance of PtSn₄ single-crystal catalyst. a) HER polarization curves for the Cu wire, Pt foil, 20% Pt/C, and PtSn₄ single crystal. b) Corresponding Tafel plots for the Pt foil, 20% Pt/C, and PtSn₄ single crystal in 1 M KOH. c) Overpotential of the PtSn₄ single-crystal catalyst at 10 mA cm^{−2} and some recently reported results for HER electrocatalysts. d) TOF values for the PtSn₄ single-crystal catalyst and other recently reported results for HER electrocatalysts. e) Current–time (*i*–*t*) chronoamperometric response of the PtSn₄ electrocatalyst for increasing current densities from 20–300 mA cm^{−2}, at an overpotential of 46 mV. The inset shows the changes in the current density during stirring of the electrolyte with a magnetic bar.

of 37 mV is required, which is close to that of the Pt/C nanostructure (28 mV) and much less than that of Pt foil (71 mV). It should be noted that the PtSn₄ single crystal demonstrates a higher activity than that of even pure Pt foils and Pt/C nanostructures (at higher current densities) in alkaline media, although Pt has a much smaller ΔG_{H^+} value. This may be caused by the sluggish water dissociation kinetics of Pt in alkaline media, which need not be considered under acidic conditions. Thus, we calculated the water dissociation energy barrier for the Pt (111) and PtSn₄ (010) surfaces, respectively. Indeed, as illustrated in Figure 4e, PtSn₄ has a much lower energy barrier of 0.41 eV than that for Pt (0.91 eV).

A Tafel analysis yields Tafel slopes of 39, 34, and 74 mV dec^{−1} for the PtSn₄ single crystal, 20% Pt/C, and Pt foil, respectively, implying a rapid HER rate with a Volmer-Heyrovsky pathway as the rate-determining step (Figure 2b). The calculated HER exchange current density (*j*₀) for the PtSn₄ single crystal is 0.585 mA cm^{−2}, which is greater than that of the nanostructure Pt/C catalyst (0.518 mA cm^{−2}). This indicates that PtSn₄ is a superior electrocatalyst in alkaline media, with a better performance compared with those of nanostructures such as Pt₃Ni nanowires (51 mV),^[14] Pt–Ni excavated nano-multipods (71 mV),^[15] and PtSe₂ nanofilms (327 mV)^[16] (Figure 2c and Table S5).^[17] TOF for each active site of PtSn₄ is calculated to determine the intrinsic activity (Figure S4, see the Supporting Information). The TOF values of PtSn₄ at 50 and 100 mV are 0.375 and 1.54 H₂s^{−1}, respectively, which are greater than those of Pt/C (0.27 H₂s^{−1} @ 50 mV) and pure Au–Ru nanowires (0.31 H₂s^{−1} @ 50 mV),^[18] and much greater than those of

other nanostructured electrocatalysts (Figure 2d).^[19] The multiple-step chrono potentiometric process employed for the PtSn₄ electrode is displayed in Figure S5, where the current density is increased from 10 to 110 mA cm^{−2}, without an *iR* correction. Simultaneous changes in the current densities and applied voltage demonstrated excellent mass transport, conductivity, and mechanical robustness of the single crystalline electrode.^[20]

We further investigated the long-term electrochemical and chemical stability of the single-crystal electrode. The PtSn₄ bulk single crystal shows excellent stability for an increasing current of 20 to 300 mA cm^{−2}, with a total measurement time of 175 h. At a constant overpotential of 46 mV, the current density is maintained for approximately 400 h without significant activity loss (Figure 2e). Moreover, the HER activity can be retained even after long-time air exposure. As shown in Figure S6, the polarization curves were recorded after three-months of air exposure, and 10000 cyclic voltammetry sweeps, showing a similar performance to that of the initial tests.

To better understand the origin of the HER activity, the structure of the single crystal was determined by single-crystal X-ray diffraction (Table S1–S3). The structure was solved in the centrosymmetric space group *Ccca* (No. 68), with lattice parameters *a* = 6.4208(7), *b* = 11.3592(1), and *c* = 6.3766(7) Å at 300 K. Figures 3a and S7 demonstrate the layered structure of PtSn₄, which is composed of stacked Sn–Pt–Sn layers along the *b* axis with a Pt–Sn distance of 2.7905(9) Å (selected crystallographic data are provided in the Supporting Information, CCDC 1841795 contain the supplementary crystallographic data for this paper. These data can be obtained free of charge from The Cambridge Crystallographic Data Centre.). A homogeneous composition and orientation can be observed from the uniform contrast of the backscatter electron images (Figure 3b and Figure S8). A thin lamella sample for high-resolution transmission electron microscopy (TEM) was fabricated by focused ion beam (FIB). Clear lattice fringes with an interplane distance of 0.32 nm, which correspond to the (200) plane of PtSn₄ can be seen from the lattice-resolved TEM image (Figure S9a,b). The high quality of the single crystal was further demonstrated by an excellent match between the experimental and simulated electron-diffraction patterns along the [010] direction (see the inset in Figure 3c and Figure S10). The atomic structure of the PtSn₄ single crystal was further investigated with noise corrected high-resolution scanning TEM (HR-STEM) (Figure 3c). The Sn–Pt–Sn stacking sequence was retained, with a 100% occupancy of the Pt and Sn sites.

The in-plane resistivity (ρ) of the PtSn₄ crystal was measured in the temperature range 2–300 K, which displays a linear temperature dependence above 50 K (Figure 3d), reflecting the dominance of the acoustic phonon scattering

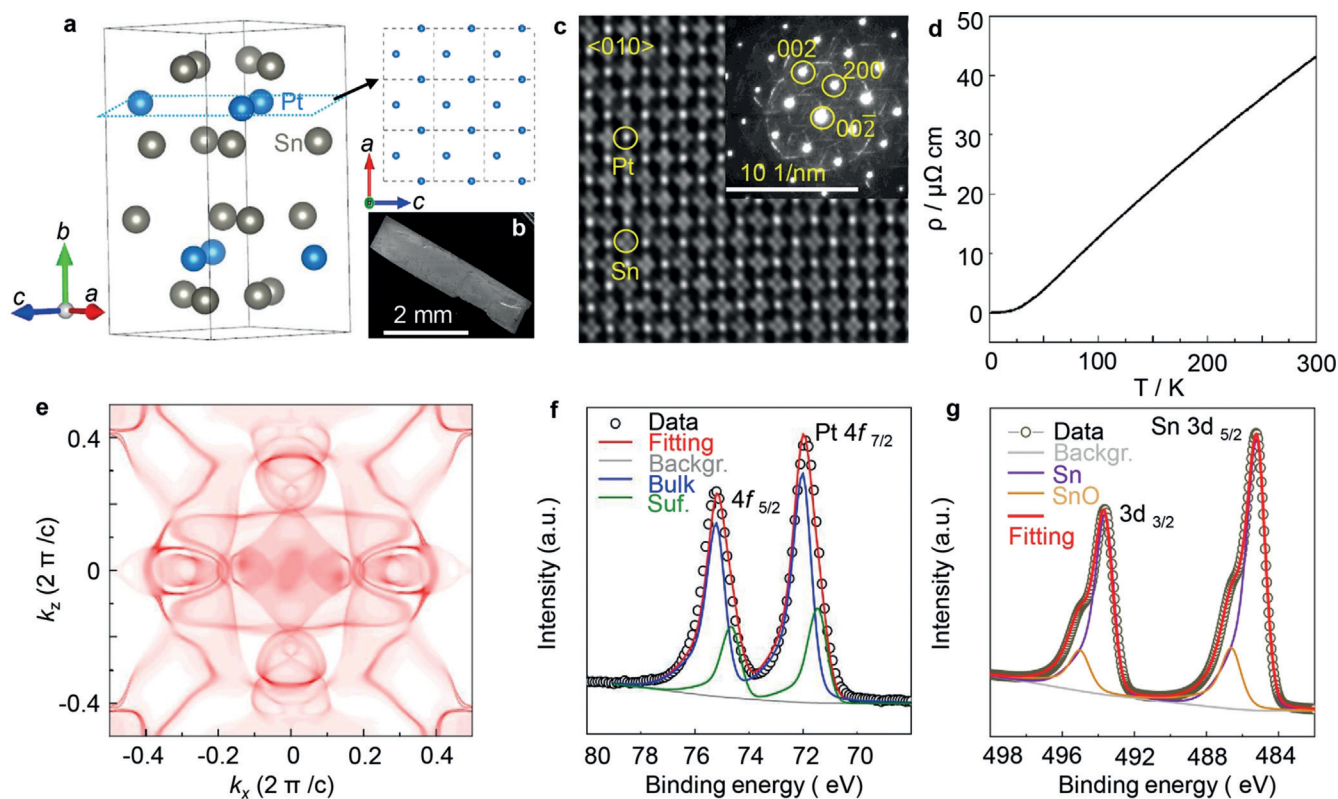


Figure 3. Structure information of bulk single-crystal PtSn_4 . a) Crystal structure of the PtSn_4 derived from the single-crystal XRD at 300 K. The exposed Pt atomic layer was constructed as shown by the upper left panel. b) A typical SEM image of the obtained PtSn_4 single crystal. c) HRTEM image of the PtSn_4 sample prepared using the focused ion beam technique (FIB). Inset: the selected area diffraction (SAED) pattern recorded along the $[010]$ crystal orientation. d) Temperature-dependent resistivity along the $[010]$ direction. An ultralow ρ of $42 \mu\Omega\text{cm}$ at room temperature was observed. e) Calculated surface Fermi surface based on the semi-infinite model. The XPS results for the f) Pt $4f$ and g) Sn $3d$.

over the electron transport. The resistivity exhibits an ultralow value of $\rho \approx 42 \mu\Omega\text{cm}$ at room temperature, which is as low as the values of pure Pt and Sn. A very large residual resistivity ratio $RRR = \rho(300 \text{ K}) / \rho(2 \text{ K}) = 1000$ was obtained, together with a non-saturating quadratic magneto-resistivity up to 9 T (Figure S11a), suggesting the high quality of the single crystal. The positive slope of the Hall resistivity as a function of applied magnetic field, as well as the positive thermopower (Figure S11b), indicate that the holes were the dominant charge carriers. This ultrahigh conductivity and mobility may be derived from the unique Fermi surface and band dispersion of the PtSn_4 single-crystal, which is a general characteristic of nodal arc semimetals. Indeed, our band structure calculation showed a band crossing just below the Fermi level (Figure S12). The (010) face Fermi surfaces were constructed as shown in Figure 3e, which consist of large electron and hole pockets.^[9c]

The electronic state of the elements on the PtSn_4 single-crystal surface was determined by X-ray photoelectron spectroscopy (XPS). The XPS survey spectrum confirmed the presence of Pt and Sn (Figure S13). As shown in Figure 3f, the main peak at a binding energy of 72.0 eV can be assigned to zero-valence platinum. It is interesting to note that no Pt-O oxidation states were observed (72.8 eV for Pt^{2+} and 74 eV for Pt^{4+}).^[21] The peak at lower binding energy originates from the surface Pt in PtSn_4 single crystal. This confirms the ultra-high stability of the Pt sites in the crystal.

The main peak with a binding energy of 485.2 eV can be attributed to $\text{Sn}^0 3d_{5/2}$.^[22] All of these results are consistent with the previous studies on Pt single crystal surface and Pt alloys, where it was claimed that the formation of Pt-O bonding is very difficult.^[23]

To understand the bonding character, the electron density distribution and electron localization function (ELF) were calculated. The total charge density distribution map is shown in Figure 4a,b, demonstrating that the electronic charge is almost completely distributed in the vicinity of the Pt and Sn atoms. This indicates the presence of a metallic bond character in the compound.^[24] The ELF iso-surface value of 0.72 further highlights the metallic nature of the bonding between Pt and Sn, and suggests a certain amount of electron localization between the intralayer Sn atoms (Figure 4c). The relatively high ELF value between the Sn atoms indicates the weak covalent nature of the Sn-Sn bonds. Finally, to confirm the exposure of the Pt layer, scanning tunneling microscopy (STM) was used to examine the in situ surface structure of the PtSn_4 (010) surface (Figure S14a). The atomically resolved image in Figure 4d visualizes a square Pt (010) net with an interatomic distance of 0.45 nm at various points in the spirals, which is identical to the bulk orientation and atomic model of the PtSn_4 (010) face, and in perfect agreement with the value obtained by single-crystal X-ray diffraction. This is further demonstrated by the significant difference in the ΔG_{H^*} values of the Sn and Pt layers (Figure 4e).

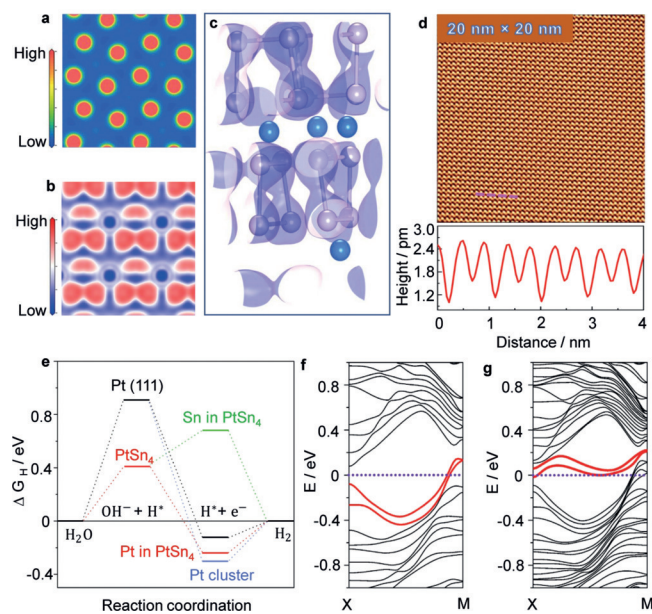


Figure 4. Bonding information in the crystal and the electron transfer mechanism. a) Contour plots of the total charge distribution of PtSn_4 in the (010) plane. The electronic charges are almost all distributed in the vicinity of the Pt and Sn atoms. b) The 3D-electron localization function mapping of the (010) surface shows that there is a weak bonding between the in-plane Sn atoms. c) The 3D-electron localization function (ELF), with an iso-surface value of 0.72. d) Atomically resolved STM topography of the PtSn_4 (010) surface. Inset: the corresponding fast Fourier transform (FFT). The line profile measured along the pink line in (d) indicates a lattice parameter of 0.451 nm; the bias voltage was $V_b = 100$ mV with a tunnel current of $I_{\text{set}} = 200$ pA. e) The Gibbs free energy diagram of the PtSn_4 (010) surface with an exposed Sn layer, Pt layer, Pt metal (111) face, and Pt nanocluster (55 atoms) calculated at the equilibrium potential of different models, and the water dissociation energy barriers on the Pt (111) and PtSn_4 (010) surfaces. f) Band structure of the PtSn_4 (010) surface with an exposed Pt layer, but without H adsorption. The robust surface states are along the direction $X \rightarrow M$ in the Brillouin zone, located just below the Fermi energy. g) Evolution of the Pt derived surface states after hydrogen adsorption when exposing the Pt layer. The surface states donate electrons to the adsorbed H 1s orbital, and shift above the Fermi level.

The ultrahigh chemical stability and defined crystal surface of PtSn_4 provides a good opportunity to investigate the related activity mechanism. DFT calculations were performed on the exposed (010) surface of the PtSn_4 single crystal. First, the exposure of Sn atoms was considered (the computation details are provided in Supporting Information). Figure S15 displays the calculated surface band structure for the (010) surface with an exposed Sn layer. The abrupt breaking of the weak Sn–Sn bonding creates a Sn-derived occupied *s*-type surface state that typically covers a broad energy range of the surface Brillouin zone. After hydrogen adsorption, the surface states corresponding to the Sn–H occupied bonding states are submerged into the bulk valence bands (Figure S16), with the H 1s orbital observed in the deep bulk band (Figure S17). However, when the Pt layer is exposed, the surfaces could not be passivated or eliminated by hydrogen as there is no Fermi level pinning at the crystal surface. Indeed, as shown in Figure S18, we observe surface

states along the X–M axis of the surface Brillouin zone (Figures 4 f and Figure S18). The orbital analysis showed that the surface states are mainly formed of p-like electronic states from the topmost Pt layer (Figures S19a and S19b). To investigate their interaction with hydrogen, the band structures of the hydrogen-covered surface were also determined. The hydrogen adsorption shifts the surface states above the Fermi level (Figure 4 g and Figure S20). This indicates the transfer of electrons from the surface state to the hydrogen *s*-like orbitals. The downshift of hydrogen bands to a position far below the Fermi energy provided further proof of this electron transfer (Figure S21).

Conclusion

The bulk single-crystal PtSn_4 exhibits Pt-like electrocatalytic activity as a catalyst for HERs. The ultra-high chemical and electrochemical stability presented by this compound, together with a significant decrease in Pt usage, makes it an ideal candidate to replace Pt catalysts. The linear band crossing in this semimetal provides large room-temperature carrier mobility and conductivity, which can be attributed to the high Fermi velocity of massless Dirac states.^[25] This facilitates the rapid charger transfer in the catalysis process and thereby increases the HER kinetics. Intrinsic and stable surface states were found in the Pt layer of the PtSn_4 single crystal. This led to the direct donation of electrons from the occupied robust surface states to adsorbed hydrogen. Rather than adopting the traditional approaches of creating active sites by nanosizing, doping, straining, and edging, the present study indicates that the manipulation of robust surface states, as well as providing high conductivity and mobility, may be used to provide a new guide for obtaining high-performance catalysts.

Acknowledgements

We gratefully acknowledge Dr. Qiunan Xu and Dr. Yang Zhang for helpful discussions. This work was financially supported by the European Research Council (ERC Advanced Grant No. 291472 “Idea Heusler”) and ERC Advanced Grant (No. 742068). TOPMAT. The calculations were carried out at the HPC Platform of ShanghaiTech University.

Conflict of interest

The authors declare no conflict of interest.

Keywords: Dirac semimetal · electrocatalysis · hydrogen evolution reaction · PtSn_4 · surface states

How to cite: *Angew. Chem. Int. Ed.* **2019**, *58*, 13107–13112
Angew. Chem. **2019**, *131*, 13241–13246

- [1] a) C. N. R. Rao, S. Dey, *Proc. Natl. Acad. Sci. USA* **2017**, *114*, 13385–13393; b) K. Sivula, R. Zboril, F. Le Formal, R. Robert, A. Weidenkaff, J. Tucek, J. Frydrych, M. Gratzel, *J. Am. Chem. Soc.* **2010**, *132*, 7436–7444; c) J. Miao, F. X. Xiao, H. B. Yang, S. Y. Khoo, J. Chen, Z. Fan, Y. Y. Hsu, H. M. Chen, H. Zhang, B. Liu, *Sci. Adv.* **2015**, *1*, e1500259; d) V. Stamenkovic, B. S. Mun, K. J. Mayrhofer, P. N. Ross, N. M. Markovic, J. Rossmeisl, J. Greeley, J. K. Norskov, *Angew. Chem. Int. Ed.* **2006**, *45*, 2897–2901; *Angew. Chem.* **2006**, *118*, 2963–2967.
- [2] a) J. Kibsgaard, T. F. Jaramillo, F. Besenbacher, *Nat. Chem.* **2014**, *6*, 248–253; b) Z. Chen, D. Cummins, B. N. Reinecke, E. Clark, M. K. Sunkara, T. F. Jaramillo, *Nano Lett.* **2011**, *11*, 4168–4175; c) G. Li, Y. Sun, J. Rao, J. Wu, A. Kumar, Q. N. Xu, C. Fu, E. Liu, G. R. Blake, P. Werner, B. Shao, K. Liu, S. Parkin, X. Liu, M. Fahlman, S.-C. Liou, G. Auffermann, J. Zhang, C. Felser, X. Feng, *Adv. Energy Mater.* **2018**, *8*, 1801258.
- [3] a) Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Norskov, T. F. Jaramillo, *Science* **2017**, *355*, eaad4998; b) Z. Pu, I. S. Amiinu, Z. Kou, W. Li, S. Mu, *Angew. Chem. Int. Ed.* **2017**, *56*, 11559–11564; *Angew. Chem.* **2017**, *129*, 11717–11722.
- [4] a) B. Yan, B. Stadtmüller, N. Haag, S. Jakobs, J. Seidel, D. Jungkenn, S. Mathias, M. Cinchetti, M. Aeschlimann, C. Felser, *Nat. Commun.* **2015**, *6*, 10167; b) J. Wiebe, F. Meier, K. Hashimoto, G. Bihlmayer, S. Blügel, P. Ferriani, S. Heinze, R. Wiesendanger, *Phys. Rev. B* **2005**, *72*, 193406.
- [5] C. Chen, S. He, H. Weng, W. Zhang, L. Zhao, H. Liu, X. Jia, D. Mou, S. Liu, J. He, Y. Peng, Y. Feng, Z. Xie, G. Liu, X. Dong, J. Zhang, X. Wang, Q. Peng, Z. Wang, S. Zhang, F. Yang, C. Chen, Z. Xu, X. Dai, Z. Fang, X. J. Zhou, *Proc. Natl. Acad. Sci. USA* **2012**, *109*, 3694–3698.
- [6] A. Politano, G. Chiarello, Z. Li, V. Fabio, L. Wang, L. Guo, X. Chen, D. W. Boukhvalov, *Adv. Funct. Mater.* **2018**, *28*, 1800511.
- [7] C. R. Rajamathi, U. Gupta, K. Pal, N. Kumar, H. Yang, Y. Sun, C. Shekhar, B. Yan, S. Parkin, U. V. Waghmare, C. Felser, C. N. R. Rao, *ChemPhysChem* **2017**, *18*, 2322–2327.
- [8] C. R. Rajamathi, U. Gupta, N. Kumar, H. Yang, Y. Sun, V. Süß, C. Shekhar, M. Schmidt, B. Yan, S. Parkin, C. Felser, C. N. R. Rao, *Adv. Mater.* **2017**, *29*, 1606202.
- [9] a) J. Li, H. Ma, Q. Xie, S. Feng, S. Ullah, R. Li, J. Dong, D. Li, Y. Li, X.-Q. Chen, *Sci. China Mater.* **2017**, *61*, 23–29; b) L.-L. Wang, A. Kaminski, P. C. Canfield, D. D. Johnson, *J. Phys. Chem. C* **2018**, *122*, 705–713; c) Y. Wu, L.-L. Wang, E. Mun, D. D. Johnson, D. Mou, L. Huang, Y. Lee, S. L. Bud'ko, P. C. Canfield, A. Kaminski, *Nat. Phys.* **2016**, *12*, 667–671.
- [10] a) Y. J. Wang, D. D. Liang, M. Ge, J. Yang, J. X. Gong, L. Luo, L. Pi, W. K. Zhu, C. J. Zhang, Y. H. Zhang, *J. Phys. Condens. Matter* **2018**, *30*, 155701; b) N. H. Jo, Y. Wu, L.-L. Wang, P. P. Orth, S. S. Downing, S. Manni, D. Mou, D. D. Johnson, A. Kaminski, S. L. Bud'ko, P. C. Canfield, *Phys. Rev. B* **2017**, *96*, 165145.
- [11] a) Y. Jiao, Y. Zheng, K. Davey, S.-Z. Qiao, *Nat. Energy* **2016**, *1*, 16130; b) J. Greeley, T. F. Jaramillo, J. Bonde, I. B. Chorkendorff, J. K. Norskov, *Nat. Mater.* **2006**, *5*, 909–913.
- [12] J. Su, Y. Yang, G. Xia, J. Chen, P. Jiang, Q. Chen, *Nat. Commun.* **2017**, *8*, 14969.
- [13] a) C. Fu, T. Scaffidi, J. Waissman, Y. Sun, R. Saha, S. J. Watzman, A. K. Srivastava, G. Li, W. Schnelle, P. Werner, M. E. Kamminga, S. Sachdev, S. S. P. Parkin, S. A. Hartnoll, C. Felser, J. Gooth, *arXiv:1802.09468v1 [cond-mat.mtrl-sci]* **2018**; b) L. Li, Y. Yu, G. J. Ye, Q. Ge, X. Ou, H. Wu, D. Feng, X. H. Chen, Y. Zhang, *Nat. Nanotechnol.* **2014**, *9*, 372–377; c) Y. Zhao, J. Qiao, Z. Yu, P. Yu, K. Xu, S. P. Lau, W. Zhou, Z. Liu, X. Wang, W. Ji, Y. Chai, *Adv. Mater.* **2017**, *29*, 1604230.
- [14] P. Wang, X. Zhang, J. Zhang, S. Wan, S. Guo, G. Lu, J. Yao, X. Huang, *Nat. Commun.* **2017**, *8*, 14580.
- [15] Z. Cao, Q. Chen, J. Zhang, H. Li, Y. Jiang, S. Shen, G. Fu, B. A. Lu, Z. Xie, L. Zheng, *Nat. Commun.* **2017**, *8*, 15131.
- [16] S. Lin, Y. Liu, Z. Hu, W. Lu, C. H. Mak, L. Zeng, J. Zhao, Y. Li, F. Yan, Y. H. Tsang, X. Zhang, S. P. Lau, *Nano Energy* **2017**, *42*, 26–33.
- [17] Y. Zheng, Y. Jiao, Y. Zhu, L. H. Li, Y. Han, Y. Chen, M. Jaroniec, S. Z. Qiao, *J. Am. Chem. Soc.* **2016**, *138*, 16174–16181.
- [18] Q. Lu, A. L. Wang, Y. Gong, W. Hao, H. Cheng, J. Chen, B. Li, N. Yang, W. Niu, J. Wang, Y. Yu, X. Zhang, Y. Chen, Z. Fan, X. J. Wu, J. Chen, J. Luo, S. Li, L. Gu, H. Zhang, *Nat. Chem.* **2018**, *10*, 456.
- [19] a) J. R. McKone, B. F. Sadler, C. A. Werlang, N. S. Lewis, H. B. Gray, *ACS Catal.* **2013**, *3*, 166–169; b) R. Zhang, X. Wang, S. Yu, T. Wen, X. Zhu, F. Yang, X. Sun, X. Wang, W. Hu, *Adv. Mater.* **2017**, *29*, 1605520; c) T. Ling, D. Y. Yan, H. Wang, Y. Jiao, Z. Hu, Y. Zheng, L. Zheng, J. Mao, H. Liu, X. W. Du, M. Jaroniec, S. Z. Qiao, *Nat. Commun.* **2017**, *8*, 1509; d) J. Kibsgaard, T. F. Jaramillo, *Angew. Chem. Int. Ed.* **2014**, *53*, 14433–14437; *Angew. Chem.* **2014**, *126*, 14661–14665.
- [20] X. Lu, C. Zhao, *Nat. Commun.* **2015**, *6*, 6616.
- [21] J. Liu, M. Jiao, L. Lu, H. M. Barkholtz, Y. Li, Y. Wang, L. Jiang, Z. Wu, D. J. Liu, L. Zhuang, C. Ma, J. Zeng, B. Zhang, D. Su, P. Song, W. Xing, W. Xu, Y. Wang, Z. Jiang, G. Sun, *Nat. Commun.* **2017**, *8*, 15938.
- [22] Y. Okamoto, W. J. Carter, D. M. Hercules, *Appl. Spectrosc.* **1979**, *33*, 287–293.
- [23] a) R. C. Baetzold, G. Apai, E. Shustorovich, R. Jaeger, *Phys. Rev. B* **1982**, *26*, 4022–4027; b) L. Zhao, J. P. Thomas, N. F. Heinig, M. Abd-Ellah, X. Wang, K. T. Leung, *J. Mater. Chem. C* **2014**, *2*, 2707.
- [24] E. Mun, H. Ko, G. J. Miller, G. D. Samolyuk, S. L. Bud'ko, P. C. Canfield, *Phys. Rev. B* **2012**, *85*, 035135.
- [25] C. Shekhar, A. K. Nayak, Y. Sun, M. Schmidt, M. Nicklas, I. Leermakers, U. Zeitler, Y. Skourski, J. Wosnitza, Z. Liu, Y. Chen, W. Schnelle, H. Borrmann, Y. Grin, C. Felser, B. Yan, *Nat. Phys.* **2015**, *11*, 645–649.

Manuscript received: May 16, 2019

Accepted manuscript online: July 25, 2019

Version of record online: August 13, 2019